

## Charge–Discharge Properties of Polypyrrole Films Containing Manganese Dioxide Particles

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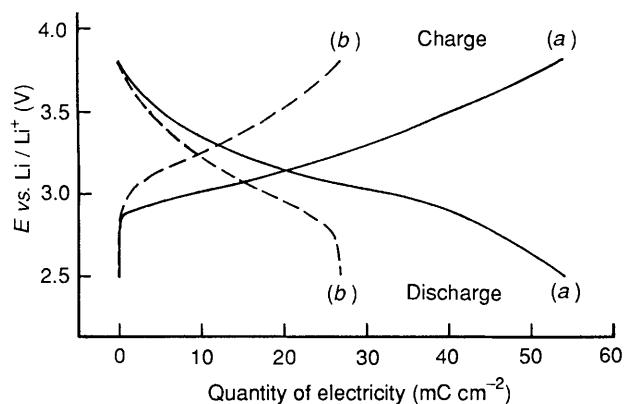
Electrochemical preparation of  $\beta$ -MnO<sub>2</sub>-incorporated polypyrrole films resulted in high energy capacity cathode-active materials for rechargeable lithium batteries.

Recently, intensive studies have been carried out on the use of conducting polymers as cathode-active materials of rechargeable lithium batteries.<sup>1–5</sup> The charge–discharge capacity of the polymer films is limited by the maximum amount of electrolyte anions incorporated into oxidized films, which have been reported to be *ca.* 0.3 and 0.5 moles per mole of monomer unit of polypyrrole<sup>1</sup> and polyaniline,<sup>5</sup> respectively; these values are not large enough to construct high energy density Li batteries. In the construction of high energy density Li batteries, the energy capacity per unit volume rather than the energy capacity per unit weight is important because of the limitation of the cell volume. This is another serious problem in the use of conducting polymers as the active materials for the Li batteries. Here we report the preparation of polypyrrole (PPy) films containing  $\beta$ -MnO<sub>2</sub> particles which allow the redox reactions of both PPy and the incorporated MnO<sub>2</sub> particles and improve the low energy capacity per unit volume of PPy.

Electrolysis at 0.1 mA cm<sup>-2</sup> of a propylene carbonate solution containing pyrrole (0.1 mol dm<sup>-3</sup>), tetraethylammonium chloride (0.5 mmol dm<sup>-3</sup>) and various amounts of suspended  $\beta$ -MnO<sub>2</sub> yielded  $\beta$ -MnO<sub>2</sub>-incorporated polypyrrole films (PPy–MnO<sub>2</sub>) on a glassy carbon anode. The  $\beta$ -MnO<sub>2</sub> particles used had a specific surface area of 23.3 m<sup>2</sup> g<sup>-1</sup>. X-Ray microanalysis revealed that the MnO<sub>2</sub> particles were distributed uniformly in the films. In our previous study on the incorporation of TiO<sub>2</sub> into PPy films in aqueous solutions containing small amounts of Na<sub>2</sub>SO<sub>4</sub> or NaI,<sup>6</sup> Cl<sup>-</sup> anion-adsorbed MnO<sub>2</sub> must be incorporated into the PPy film as a

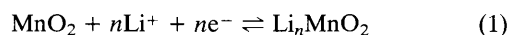
charge compensator for oxidized PPy produced by the electrochemical oxidation of pyrrole. The amount of the incorporated MnO<sub>2</sub> was determined by first firing the PPy films at 1000 °C, then dissolving the resulting MnO<sub>2</sub> in a boiling acid mixture consisting of 10% H<sub>2</sub>SO<sub>4</sub>, 12% H<sub>3</sub>PO<sub>4</sub>, 17% HNO<sub>3</sub> and 5% KIO<sub>4</sub>, and finally by applying absorption spectrometry to the resulting solution at 525 nm. PPy–MnO<sub>2</sub> prepared with the deposition charge ( $Q_d$ ) of 900 mC cm<sup>-2</sup> from deposition baths containing 5, 10, 20 and 40 g dm<sup>-3</sup> of MnO<sub>2</sub> contained 40.7, 50.4, 49.7 and 50.2 wt% of MnO<sub>2</sub> particles, respectively, suggesting that the amount of the incorporated MnO<sub>2</sub> in the PPy film became constant for suspended MnO<sub>2</sub> greater than 10 g dm<sup>-3</sup>. If  $Q_d$  of PPy–MnO<sub>2</sub> was varied, the amount of MnO<sub>2</sub> incorporation seemed to change, depending on  $Q_d$ , as judged from rough proportionality between  $Q_d$  and the charge–discharge capacity of the resulting PPy–MnO<sub>2</sub>, which will be described in a later section.

The charge–discharge test of the prepared PPy films was carried out in a mixed solution of propylene carbonate and dimethoxyethane (1:1) containing LiClO<sub>4</sub> (1 mol dm<sup>-3</sup>). Figure 1 shows the charge–discharge profiles of the PPy–MnO<sub>2</sub> and MnO<sub>2</sub>-free PPy films taken at 0.1 mA cm<sup>-2</sup> in a potential range between 2.5 and 3.8 V *vs.* Li/Li<sup>+</sup>. The charge–discharge capacity of MnO<sub>2</sub>-free PPy film was 26.6 mC cm<sup>-2</sup>, while that of the PPy–MnO<sub>2</sub> was 54 mC cm<sup>-2</sup>. X-Ray diffraction analyses were carried out for oxidized and reduced forms of the PPy–MnO<sub>2</sub> films prepared by polarizing the film at 3.8 and 2.5 V *vs.* Li/Li<sup>+</sup>, respectively, for 30 min in the



**Fig. 1** Charge-discharge curves of polypyrrole film containing 50.2 wt%  $\beta$ - $\text{MnO}_2$  particles (a) and  $\text{MnO}_2$ -free PPy film (b) taken at  $0.2 \text{ mA cm}^{-2}$  in a mixed solution of propylene carbonate and dimethoxyethane (1 : 1) containing  $\text{LiClO}_4$  ( $1 \text{ mol dm}^{-3}$ ).  $\beta$ - $\text{MnO}_2$ -incorporated polypyrrole film was prepared using a coating bath containing  $40 \text{ g dm}^{-3}$   $\text{MnO}_2$ . The quantity of electricity used for the polypyrrole preparation was  $200 \text{ mC cm}^{-2}$  for both PPy- $\text{MnO}_2$  and  $\text{MnO}_2$ -free PPy films.

above-mentioned electrolyte solution, followed by vacuum drying for 5 h. Sharp diffraction peaks due to (100), (101) and (211) phases of the  $\beta$ - $\text{MnO}_2$  crystals were observed for both films, but these diffraction peaks were shifted as much as ca.  $0.7^\circ$  to lower diffraction angles when the oxidized film was reduced, suggesting that the  $\beta$ - $\text{MnO}_2$  lattice was expanded with reduction of the film due to intercalation of electrolyte cations into the  $\text{MnO}_2$  lattice. Since the shift in the diffraction angle was reversible in the charge-discharge cycles, the redox reaction detailed in eqn. (1)<sup>7</sup> seemed to be valid in the  $\text{MnO}_2$ -incorporated PPy films.



Considering that two electrons are involved in the oxidation of a pyrrole ring to be polymerized and the resulting polymers contain fractional charges  $m$  per unit pyrrole ring, the theoretical capacity ( $Q_c$ ) is correlated to the deposition charge ( $Q_d$ ) as given by  $Q_c = Q_d \times m/(2 + m)$ . By inserting  $Q_d = 200 \text{ mC cm}^{-2}$  and  $m = 0.31$ , which was obtained by elemental analysis of the as-grown  $\text{MnO}_2$ -free films, into this equation,  $26.8 \text{ mC cm}^{-2}$  was obtained for  $Q_c$ . In the case of  $\text{MnO}_2$ -free PPy films, there was fair agreement between the theoretical  $Q_c$  and the experimentally obtained one ( $26.6 \text{ mC cm}^{-2}$ ). If it is assumed that the capacity due to the redox reaction of polypyrrole of the PPy- $\text{MnO}_2$  films is the same as that of  $\text{MnO}_2$ -free PPy films, then the excess capacity evaluated from the difference in the experimentally determined  $Q_c$  and the theoretical one is attributable to the charges involved in the redox reaction of the incorporated  $\text{MnO}_2$  particles. It was found that the amount of N and  $\text{MnO}_2$  in the PPy- $\text{MnO}_2$  film was 7.55 and 50.2 wt%, respectively, which gave 1.07 of the molar ratio of  $\text{MnO}_2$  to the pyrrole ring. Since the molar ratio of  $\text{Cl}^-$  ions incorporated in the PPy- $\text{MnO}_2$  film to the pyrrole ring was found to be 0.31, the number of moles of pyrrole rings ( $M_{\text{Py}}$ ) in the film was evaluated from the equation given by  $M_{\text{Py}} = Q_d/F \times 1/(2 + 0.31)$ , where  $F$  is Faraday constant. By inserting  $Q_d = 200 \text{ mC cm}^{-2}$  into the equation,  $M_{\text{Py}}$  of  $8.97 \times 10^{-7} \text{ mol cm}^{-2}$  was obtained, from which  $9.6 \times 10^{-7} \text{ mol cm}^{-2}$  was obtained for the amount of  $\text{MnO}_2$  incorporated in the film. Accordingly, the utilization of the incorporated  $\text{MnO}_2$  was 0.30.

Table 1 gives the results of charge-discharge tests of the PPy and the PPy- $\text{MnO}_2$  films, the film thickness determined by observation of the cross-section of the films using a scanning

**Table 1** Charge-discharge test of polypyrrole films containing  $\beta$ - $\text{MnO}_2$  particles<sup>a</sup>

Film	$Q_d^b/$ $\text{mC cm}^{-2}$	Charge- discharge capacity/ $\text{mC cm}^{-2}$	Utilization of $\text{MnO}_2$	Film thickness/ $\mu\text{m}$	Energy capacity per unit volume/ $\text{C cm}^{-3}$
PPy	200	26.6	—	1.0	266
PPy- $\text{MnO}_2$	100	26.5	0.29	0.6	441
	200	54.0	0.30	1.2	450
	500	140.5	0.31	3.1	453

<sup>a</sup> The films were prepared from propylene carbonate solution containing pyrrole ( $0.1 \text{ mol dm}^{-3}$ ), tetraethylammonium chloride ( $0.5 \text{ mmol dm}^{-3}$ ) and  $40 \text{ g dm}^{-3}$  of  $\beta$ - $\text{MnO}_2$  particles with stirring of the solution. <sup>b</sup> The quantity of electricity used for preparation of the film.

electron microscope (SEM) and the energy capacity per unit volume of the film, which was evaluated from the charge-discharge capacity and the film thickness. The surface of the PPy- $\text{MnO}_2$  was found to be fairly smooth with unevenness of  $\pm 0.1 \mu\text{m}$ . As shown in Table 1, the film thickness was increased by only 20% by incorporating ca. 50 wt%  $\text{MnO}_2$  particles into the polypyrrole film, while the increase in the capacity of the PPy- $\text{MnO}_2$  film was almost double that of the  $\text{MnO}_2$ -free polypyrrole film, as described above. Therefore, the energy capacity per unit volume of the PPy- $\text{MnO}_2$  film was 1.70 times that of the  $\text{MnO}_2$ -free PPy film, indicating that the incorporation of  $\text{MnO}_2$  particles into the film is effective for improving the energy capacity per unit volume of the PPy film. The weight of the PPy- $\text{MnO}_2$  and  $\text{MnO}_2$ -free PPy films prepared with  $Q_d = 200 \text{ mC cm}^{-2}$  was estimated to be 151.6 and  $68.2 \mu\text{g}$ , respectively, from the above-mentioned molar ratio of  $\text{Cl}^-$  to pyrrole rings, the number of moles of pyrrole rings and  $\text{MnO}_2$  in the film. Then, the energy capacity per unit weight of the PPy- $\text{MnO}_2$  and the  $\text{MnO}_2$ -free PPy films was estimated to be 350 and  $390 \text{ C g}^{-1}$ , respectively, indicating that the incorporation of  $\beta$ - $\text{MnO}_2$  particles caused a small decrease in the energy capacity per unit weight of polypyrrole.

Table 1 shows that the film thickness and the charge-discharge capacity of the PPy- $\text{MnO}_2$  film were roughly proportional to  $Q_d$ , indicating that the PPy matrix worked well as conducting networks for charge-transfer between the electrode substrate and  $\text{MnO}_2$  particles in the film. In the use of  $\text{MnO}_2$  as an active material for batteries, carbon powders like acetylene black are mixed with  $\text{MnO}_2$  to make the conducting networks for  $\text{MnO}_2$ ; the carbon in this case does not work as the active material. In contrast polypyrrole in PPy- $\text{MnO}_2$  works not only as the conducting networks but also as active materials, as already shown above. From this viewpoint too, the PPy- $\text{MnO}_2$  is superior to conventional  $\text{MnO}_2$ -cathodes as the active materials in batteries.

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